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1. INTRODUCTION

Self-processing photopolymer materials enjoy widespread use as holographic recording media. Typically, illuminating light activates a dye, initiating polymerization, which drives a diffusion-based process.\textsuperscript{1-7} The polymer chains grow away from their initiation points, removing monomers where they grow. It was recently proposed\textsuperscript{8} that one can model this situation by introducing a nonlocal response function into the one-dimensional diffusion equation\textsuperscript{9} that governs grating formation.\textsuperscript{10,11} The physical extent of the nonlocal response is indicated by a normalized nonlocal variance $\sigma$, where $\sigma = 0$ and $\sigma > 0$ correspond to the local and the nonlocal cases, respectively. The variance quantifies the effects of polymer chain growth initiated at one point in the medium on the polymerization that takes place some distance away. The nonlocal diffusion model predicts the existence of both a high- and a low-frequency cutoff in the material’s spatial frequency response.

In Refs. 8, 10, and 11 it was assumed that the change in monomer concentration is directly proportional to the recording interference pattern’s intensity. An extension of the local diffusion model\textsuperscript{12} uses a general relationship between the illuminating intensity and the polymerization rate. In a recent paper, starting with the physical analysis of Odian,\textsuperscript{13} Kwon \textit{et al.}\textsuperscript{14} derived a rate of polymerization that is directly proportional to the square root of the illuminating intensity. In the present paper we examine the dependence of the square root on exposure intensity and incorporate it into the coupled harmonic nonlocal model. The resultant corrected model, model II, and the uncorrected model, model I, are then compared.

We derive approximate analytic solutions to the nonlocal diffusion equation, assuming a two-harmonic expansion of the monomer concentration. These analytic expressions are compared with the more rigorous four-harmonic numerical results, indicating the range of physical parameters for which they are accurate. The predictions of the model, corrected to allow for Fresnel reflections during recording and replay, are then compared with growth curves that we produced by measuring the diffraction intensities produced by volume holographic gratings with different spatial frequencies.

The holographic gratings used were all recorded in an acrylamide-based photopolymer material.\textsuperscript{15-18} We calculated the resultant diffraction efficiencies, allowing for Fresnel boundary losses. By applying Kogelnik’s coupled-wave theory,\textsuperscript{4,7,19,20} we calculated the corresponding refractive-index modulations. Fitting these curves by using the two-harmonic analytic expressions, we estimated the bulk physical parameters that govern the behavior of the material. These parameters included the monomer diffusion constant $D$, the rate of polymerization $F_0$, the constant of proportionality between the exposure
energy and the rate of polymerization $\kappa$, the constant of proportionality between the polymer concentration and the resultant refractive index $C$, the dimensionless reaction rate $R$, and the normalized nonlocal polymer chain length distribution variance $\sigma$.\footnote{The trends in the behavior of these parameters as the spatial frequency is changed are indicated.}

### 2. MODELING PHOTOCHEMICAL PROCESSES

Photopolymer systems use a sensitizer $Z$ to initiate polymerization. The sensitizer (dye) absorbs light and moves into an excited state, $Z^*$. Typically this excited state then transfers its energy to another compound, $C$, which then becomes excited, $C^*$. Free radicals $R^*$ are then formed by the decomposition of $C^*$. The mechanism for free-radical polymerization has three stages, namely, initiation, propagation, and termination. Initiation involves the production of free radicals, which bind to a monomer to form the chain-initiation species $M_1^*$:

$$R^* + M \rightarrow M_1^*, \quad (1)$$

where $k_i$ is the initiation rate constant. The radical $M_1^*$ then propagates by bonding with other monomer molecules to form a long polymer chain:

$$M_1^* + M \rightarrow M_{1+1}^*, \quad (2)$$

where $k_p$ is the propagation rate constant. The growth of the polymer chain is stopped by termination. Termination occurs when the radical attached to the end of the growing polymer chain bonds with a free radical. This reaction produces a dead polymer, i.e., a chain with no radical attached to one end and which is therefore incapable of growing further:

$$M_{n}^* + M_{m}^* \rightarrow \text{dead polymer,} \quad (3)$$

where $k_t$ is the termination rate constant. Normally the rate of termination is higher than the rate of propagation. However, the concentration of free radicals is much lower than the concentration of monomers, so the polymer chain can grow for some time before it is terminated.

The rate of disappearance of monomers can be given by

$$-\frac{\partial c}{\partial t} = R_i + R_p, \quad (4)$$

where $c$ is the monomer concentration, $R_i$ is the initiation rate, and $R_p$ is the rate of propagation. Because there is a larger number of monomers than free radicals, it can be assumed that $R_p \gg R_i$.\footnote{As we stated above, monomers diffuse from dark regions to bright regions as the concentration of monomers is modulated by the illumination. This diffusion equation can be combined with Eq. (11) to give the monomer concentration:}

$$-\frac{\partial c}{\partial t} = R_p \quad (5)$$

The rate of polymerization can be given by

$$R_p = k_p c c_r, \quad (6)$$

where $c_r$ is the total concentration of chain radicals. The radical concentration rises exponentially at first, reaching a steady state that is maintained during the polymerization. This reaching of steady state implies that the rate of initiation, $R_i$, and the rate of termination, $R_t$, are equal, yielding

$$R_i = R_t = 2k_t c_r^2. \quad (7)$$

The expression for $c_r$ is substituted into Eq. (6) to give the rate of polymerization:

$$R_p = k_p c (R_p/2k_i)^{1/2} \times R_i^{1/2}. \quad (8)$$

Clearly the rate of propagation is proportional to the square root of the initiation rate. The initiation rate is $R_i = 2\Phi I_a$, where $\Phi$ is the quantum yield of radical production, which can be defined as the number of pairs of radicals formed per quantum of light absorbed.\footnote{The expression for $c_r$ is substituted into Eq. (6) to give the rate of polymerization:} $I_a$ is the intensity of light absorbed, expressed in moles of light quanta per liter per second. Given that the spatially modulated illumination is $I(x) = I_0[1 + V \cos(Kx)]$, where $V$ is the visibility and the grating vector magnitude is $K = 2 \pi / \Lambda$, where $\Lambda$ is the grating period, $I_a$ is

$$I_a(x) = I(x)[1 - \exp(-\varepsilon Z d)] = I(x)(1 - T), \quad (9)$$

where $\varepsilon$ is the molar absorptivity, $Z$ is the concentration of photosensitizers, $d$ is the thickness of the photopolymer layer, and $T$ is the transmittance of the layer. When the initiation step is started by the photosensitizers, the concentration of the radicals can be given by

$$c_r = \left[\frac{\Phi I(x)(1 - T)}{k_i}\right]^{1/2}, \quad (10)$$

and the polymerization rate, Eq. (6), is given by

$$R_p = k_p c \left[\frac{\Phi I(x)(1 - T)}{k_i}\right]^{1/2}. \quad (11)$$

Combining Eqs. (5) and (11) yields the rate of polymerization of the monomers:

$$-\frac{\partial c}{\partial t} = k_p c \left[\frac{\Phi I(x)(1 - T)}{k_i}\right]^{1/2}. \quad (12)$$

As we stated above, monomers diffuse from dark regions to bright regions as the concentration of monomers is modulated by the illumination. This diffusion equation can be combined with Eq. (11) to give the monomer concentration:

$$-\frac{\partial c}{\partial t} = D \frac{\partial^2 c}{\partial x^2} + Q[1 + V \cos(Kx)]^{1/2} c, \quad (13)$$

where

$$Q = k_p \left[\frac{\Phi I_0(1 - T)}{k_i}\right]^{1/2}$$

is analogous to $F_0$ as it appears in Refs. 10 and 11. Parameters $\kappa$ in Ref. 8 and $k_p[(\Phi(1 - T))/k_i]^{1/2}$ in Ref. 14 are similarly analogous.
3. NONLOCAL MATERIAL RESPONSE

In this section we incorporate the square root dependence into the nonlocal model. Recall the one-dimensional nonlocal response diffusion equation:

\[
\frac{\partial u(x, t)}{\partial t} = \frac{\partial}{\partial x} \left[ D(x, t) \frac{\partial u(x, t)}{\partial x} \right] - \int_{-\infty}^{+\infty} R(x, x')P(x', t)u(x', t)dx'.
\]

(14)

Assume a monomer concentration given by

\[
u(x', t) = \sum_{i=0}^{\infty} u_i(t)\cos(iKx')
\]

and a diffusion constant expanded as

\[
D(x, t) = \sum_{i=0}^{\infty} D_i(t)\cos(iKx).
\]

(16)

A Gaussian form of nonlocal material response function is chosen:

\[
R(x - x') = \frac{\exp[-(x - x')^2/2\sigma]}{\sqrt{2\pi\sigma}}.
\]

(17)

Previously it was assumed that the rate of polymerization was of the form

\[
F(x', t) = F_0[1 + V\cos(Kx')].
\]

(18)

Equation (18) is used in model I.

This equation must now be adjusted to take account of the proposed dependence on the square root of intensity:

\[
F(x', t) = F_0[1 + V\cos(Kx')]^{1/2}.
\]

(19)

Equation (19) is used in model II.

Based on our definition of \( Q \) we define \( F_0 = \kappa I_0^{1/2} \), where \( \kappa = k_p[\Phi(1 - T)]/k_1^{1/2} \). \( F(x, t) \) can be represented as a Fourier series:

\[
F(x', t) = F_0 \sum_{i=0}^{\infty} f_i \cos(iKx').
\]

(20)

We substitute Eqs. (15)–(17) and (20) into Eq. (14) to give the first-order nonlocal coupled equations that govern the time evolution of the harmonics of monomer concentration for any intensity-dependent rate of polymerization and the diffusion constant:

\[
\frac{d u_0(t)}{dt} = -F_0 f_0 u_0(t) - \frac{F_0}{2} \sum_{n=1}^{\infty} f_n u_n(t),
\]

(21a)

\[
\frac{d u_j(t)}{dt} \bigg|_{j>0} = -(jK)^2 D_0(t)u_j(t)
\]

\[
- j \frac{K^2 j - 1}{2} \sum_{n=1}^{\infty} nD_{j-n}(t)u_n(t) - F_0 f_0 u_j(t)
\]

\[
\times \exp[-(jK)^2/2]
\]

\[
- \frac{F_0 \exp[-(jK)^2/2]}{2} \sum_{i=0}^{\infty} f_{j+i} u_i(t)
\]

\[
+ \sum_{i=1}^{\infty} f_{j-i} u_i(t) + \sum_{i=1}^{j-1} f_{j-i} u_i(t)
\]

\[
+ j \frac{K^2 j}{2} \sum_{n=1}^{\infty} [nD_{n-j}(t)u_n(t)
\]

\[- (n + j)D_n(t)u_{n+j}(t)].
\]

(21b)

Several assumptions are now made. It is assumed that, in the Fourier series expansion of the diffusion constant, only the first two terms must be retained:

\[
D(x, t) = D \exp[-aF_0 t(\sqrt{1 - V} + \sqrt{1 + V}/2]
\]

\[
\times \left\{ \cosh \left[ aF_0 t(\sqrt{1 + V} - \sqrt{1 - V}) \right] \right. \]

\[
- \left. \sinh \left[ aF_0 t(\sqrt{1 + V} - \sqrt{1 - V}) \right] \cos(Kx) \right\}.
\]

(22)

This assumption was examined previously and was shown to be acceptable, because the first or average term dominates, except in cases of high nonlinearity. The diffusion constant is assumed to decay exponentially as a function of the polymerization rate, where \( \alpha \) is the diffusion constant’s decay parameter. We also introduce \( \xi \), which is proportional to the rate of polymerization:

\[
\xi = F_0 t = \kappa I_0^{1/2} t.
\]

(23)

We substitute from Eqs. (22) and (23) into Eqs. (21). Assuming that the monomer concentration profile is well approximated by a four-harmonic series, i.e., that \( u_{j>3}(\xi) = 0 \) and that \( f_{m>5} = 0 \), the four coupled monomer rate equations are

\[
\frac{d u_0(\xi)}{d\xi} = -f_0 u_0(\xi) - \frac{1}{2} [f_1 u_1(\xi) + f_2 u_2(\xi) + f_3 u_3(\xi)].
\]

(24a)
\[ \frac{du_1(\xi)}{d\xi} = -R \text{Ch}[\xi]u_1(\xi) - R \text{Sh}[\xi]u_2(\xi) \]
\[ - S_1 \left( f_1 u_0(\xi) + \left( f_0 + \frac{f_2}{2} \right) u_1(\xi) \right) + \frac{1}{2} \left( f_1 + f_3 \right) u_2(\xi) + \frac{1}{2} \left( f_2 + f_4 \right) u_3(\xi), \] (24b)

\[ \frac{du_2(\xi)}{d\xi} = -4R \text{Ch}[\xi]u_2(\xi) + R \text{Sh}[\xi] \left[ u_1(\xi) + 3u_3(\xi) \right] \]
\[ - S_2 \left[ f_2 u_0(\xi) + \frac{1}{2} \left( f_0 + f_2 \right) u_1(\xi) + \left( f_0 + f_4 \right) u_3(\xi) \right] + \frac{1}{2} \left( f_1 + f_5 \right) u_2(\xi) - \left( f_0 + \frac{f_6}{2} \right) u_3(\xi), \] (24c)

\[ \frac{du_3(\xi)}{d\xi} = -9R \text{Ch}[\xi]u_3(\xi) + 3R \text{Sh}[\xi]u_2(\xi) \]
\[ - S_3 \left[ f_3 u_0(\xi) + \frac{1}{2} f_2 + f_4 \right] u_1(\xi) - \frac{1}{2} \left( f_1 + f_5 \right) u_2(\xi) - \left( f_0 + \frac{f_6}{2} \right) u_3(\xi), \] (24d)

where \( \text{Ch}[\xi] = \exp\left[ -\alpha F_0 (\sqrt{1 - V} + \sqrt{1 + V})/2 \right] \times \cosh[\alpha F_0 (\sqrt{1 - V} + \sqrt{1 - V})/2], \)
\( \text{Sh}[\xi] = \exp\left[ -\alpha F_0 (\sqrt{1 - V} + \sqrt{1 - V})/2 \right] \sinh[\alpha F_0 (\sqrt{1 - V} + \sqrt{1 - V})/2]. \)

Equations (24) are solved numerically with the initial conditions \( u_0(0) = 100 \) and \( u_m(0) = 0 \) for all \( m > 0 \). When \( V = 1, f_0 = 2 \sqrt{2}/\pi \) and
\[ f_m = \frac{4 \sqrt{2}}{\pi} \left( -1 \right)^{m+1} \left( -1 + 4m^2 \right). \]

When the square root is eliminated, the equations reduce to model I and \( f_0 = 1, f_1 = V, \) and \( f_m > 1 = 0. \)

**4. ANALYTIC FORMULAS FOR MODEL II**

We derived analytic expressions for monomer and polymer concentrations, assuming that the concentration was well approximated by the zeroth and first harmonics alone, \( (u_0 \) and \( u_1 \). To do so, we assumed that \( \alpha = 0 \), which we could do because changing \( \alpha \) is equivalent to varying \( R \) (Ref. 10):

\[ \frac{du_0(\xi)}{d\xi} = -f_0 u_0(\xi) - f_1 u_1(\xi)/2, \] (25a)

\[ \frac{du_1(\xi)}{d\xi} = -S f_1 u_0(\xi) - W u_1(\xi), \] (25b)

with \( W = S (f_0 + f_3/2) + R \) and \( S = S_1 = \exp(-K^2 \sigma^2/2). \)

The analytic solutions to Eqs. (25) are of the form

\[ u_0(\xi) = \frac{100 \exp[(W + f_0)\xi/2]}{B} \left[ B \cosh(B\xi/2) + (W - f_0) \times \sinh(B\xi/2) \right], \] (26a)

\[ u_1(\xi) = \frac{100 S f_1}{B} \exp \left[ \frac{(B + W + f_0)\xi}{2} \right] \left[ \exp \left( \frac{B\xi}{2} \right) - 1 \right], \] (26b)

where \( B = [(W - f_0)^2 + 2f_3^2 S]^{1/2}. \)

The resultant concentration of polymerized monomers after an exposure of duration \( t \) is given by

\[ N(x, t) = \int_{-\infty}^{t} R(x - x')F(x', t')u(x', t')dx'dt', \] (27)

yielding the following polymerization concentration spatial-harmonic components:

\[ N_0(\xi) = \int_{0}^{\xi} \left[ f_0 u_0(\xi') + (f_1/2)u_1(\xi') \right] d\xi', \] (28a)

\[ N_1(\xi) = S \int_{0}^{\xi} \left[ f_1 u_0(\xi') + (f_0 + f_3/2)u_1(\xi') \right] d\xi'. \] (28b)

Substituting for \( u_0 \) and \( u_1 \) gives that

\[ N_0(\xi) = 100 + \frac{50}{B} \exp \left[ \frac{(W - f_0 - B)\xi}{2} \right] \left[ (W - f_0 - B) - (W - f_0 + B) \exp \left[ \frac{B\xi}{2} \right] \right], \] (29a)

\[ N_1(\xi) = \frac{400S}{B^2 - (W + f_0)^2} \left[ R + \exp \left[ \frac{(W + f_0)\xi}{2} \right] \right] \times \left[ \frac{L}{B} \sinh \left( \frac{B\xi}{2} \right) - R \cosh \left( \frac{B\xi}{2} \right) \right], \] (29b)

where \( L = (f_0 - R)R + [-f_0^2 + (f_0 + f_3/2)(2f_0 - R)]S. \)

It is assumed that the modulation of the refractive index induced during recording is linearly related to the polymer concentration. Therefore the change in cross-sectional refractive index recorded in the material is

\[ n(x, \xi) = C \sum_{i=0}^{1} N_i(\xi) \cos(iKx). \] (30)

**5. COMPARISON OF MODELS I AND II**

A numerical comparison of four-harmonic models I and II presented in Section 3 is carried out. For a consistent comparison we set \( R(1) = R(11) \), i.e., we assume that \( k_{i1}^{(i1)} \cdot k_{i1}^{(1)} \), in which case \( \xi(1) = \xi(11). \)

Therefore we assume that we are comparing two different media, which respond differently to the illuminating light.
pattern. The predicted monomer and polymer concentration amplitudes and the corresponding grating profiles were examined. The $R$ and $S$ values that were used distinguish previously identified regimes of material behavior.\textsuperscript{8,10}

Figure 1 shows the exposure-curve values of the zeroth harmonic ($u_0$) and the first harmonic ($u_1$) of monomer concentration predicted by model II (dashed curves) and by model I (solid curves). In fact, Fig. 1 shows the largest difference between the predictions (as we discuss below). The values of $u_0$ and $u_1$ predicted by model II follow the same general trends as were observed for model I\textsuperscript{8}; i.e., as $\sigma$ is increased, the monomer concentration falls off more rapidly and the amplitude of the higher harmonics is reduced.

Figure 2 shows an example of the differences in the zeroth ($N_0$) and first ($N_1$) harmonics of polymer concentration as predicted by the model I (solid curves) and by model II (dashed curves).

From Figs 1 and 2 it would appear that for $\xi > 10$ the system has effectively reached its final state, as no monomer remains to be polymerized. Therefore, fixing $\xi = 12$, in Figs. 3(a) and 3(b) we plot the numerical four-harmonic predictions for $N_1$ over the ranges $0.1 < S < 1$ and $0.01 < R < 100$ for models I and II, respectively. In both cases the strongest first-harmonic prediction occurs for a medium that has large $R = 100$ and large $S = 1$ values. This type of medium corresponds to a recording medium that is illuminated (intensity and period) such as to exhibit a high rate of diffusion (large $D$) and a low rate of polymerization (small $F_0$) and with short polymer chains (small $\sigma$). For both Figs. 3(a) and 3(b) the $N_1$ component drops to less than 10% of the maximum value (in the range) when $S < 0.2$ (for all $R$) and $R < 0.1$ (for all $S$). A comparison of the relative sizes of
the predictions of the two models shows that, as $R$ and $\sigma$ increase, the two predictions became more similar in size.

Figure 4 shows the resultant cross-sectional polymer concentration profiles for the largest differences between the two models (and between the illuminating intensity and the resultant recorded profile) as predicted by four-harmonic model I (solid curve) and four-harmonic model II (dashed curve). In this case $V = 1$ and $\xi = 10$. The model II profile predictions follow the same trend as that for four-harmonic nonlocal model I, i.e., (1) as $R$ increases, the profiles become more sinusoidal in shape, and (2) increasing $\sigma$ smooths the profile but decreases its visibility. For all values of $R$ and $\sigma$ examined, the visibility of the model II profile was found to be greater than that of the corresponding model I profile. The difference decreases as $R$ and $\sigma$ increase.

6. ACCURACY OF THE ANALYTIC FORMULAS

To quantify the accuracy of the two-harmonic model I and model II analytic formulas we define a percentage difference calculated from the formula

$$\% = \frac{N_1(4 - \text{harmonic}) - N_1(2 - \text{harmonic})}{N_1(4 - \text{harmonic})} \times 100,$$

(31)

where all steady-state values of $N$ are calculated at $\xi = 12$. The results for the two models are shown in Fig. 5. It can be seen that for $R > 1$ (and all $S$ values) and for $S < 0.2$ (and all $R$ values) the difference between the numerical four-harmonic and the analytic two-harmonic predictions are less than 5% of the more accurate numerical result.

Comparing these results with the four-harmonic prediction presented in Fig. 3 makes clear that the disagreement presented in Fig. 3 is less than 5% of the more accurate numerical result.

7. APPLICATION TO EXPERIMENTAL DATA

In this section we estimate the physical parameters of the material by fitting experimental growth curves, using models I and II.

A. Polynomial Fits

Starting with the two-harmonic analytic solution for grating-refractive-index modulation [Eq. (33b)], we find the Taylor series approximation to $n_1(\xi)$ close to $\xi = 0$, using relation (35b). The first three terms of this exact expansion are

$$
100CSf_1\xi - 25CSf_1[S(2f_0 + f_2) + 2f_0]\xi^2
+ \frac{25}{6} CSf_1(4f_0^2 + (2f_0 + f_2)^2S^2
+ 2[f_1^2 + S(2f_0 + f_2)(f_0 + R)])\xi^3
= a_1\xi + a_2\xi^2 + a_3\xi^3 + \ldots .
$$

(32)

The $n_1$ experimental growth curves are produced from measured diffraction efficiencies corrected for Fresnel reflections. In the experimental curves $n_1$ is plotted as a function of $I_0 \times t$, the exposure energy, and the polynomial fits to the experimental data are made in terms of this parameter:

$$
n_1(I_0 \times t) = e_1(I_0 \times t) + e_2(I_0 \times t)^2 + e_3(I_0 \times t)^3
+ \ldots + e_n(I_0 \times t)^n .
$$

(33a)

So the polynomial fit gives us the coefficients $e_1$, $e_2$, $e_3$, etc. In model II, however, $\xi = \kappa I_0^{1/2} t$. As we know the value of $I_0$ that was used during the fabrication of the gratings, we can compare like with like by rewriting Eq. (33a) in the form

$$
n_1(I_0^{1/2} \times t) = [e_1 \times I_0^{1/2}][I_0^{1/2} \times t] + [e_2 \times I_0][I_0^{1/2} \times t]^2
+ \ldots + [e_n \times I_0^{1/2}][I_0^{1/2} \times t]^n .
$$

(33b)

Equating the corresponding coefficients gives us that
\[ e_1 \times I_0^{2/2} \equiv a_1^1 \kappa = 100CSf_1\kappa, \]  
\[ e_2 \times I_0^{2/2} \equiv a_2^2 \kappa^2 = -25CSf_1[S(2f_0 + f_2) + 2f_0]\kappa^2, \]  
and so on.

First let us assume that the material response is local; i.e., that \( \sigma \to 0 \) and \( S \to 1 \). Then, from relations (34), \( \kappa \) can be estimated:

\[ \frac{e_2 \times I_0^{2/2}}{e_1 \times I_0^{2/2}} \equiv \frac{a_2^2 \kappa^2}{a_1^1 \kappa} = -\frac{4f_0 + f_2}{\kappa}. \]  
Therefore it is possible to determine a value for \( \kappa \), given the experimentally extracted polynomial coefficients \((a_1^1 \kappa)\) and \((a_2^2 \kappa^2)\), from the fits to the experimental \( n_1^1(I_0t) \) data curves. \( C \) can then be found, assuming that \( \kappa \) is already known, and a value for \( R \) can then be found by use of the values for \( \kappa \) and \( C \).

Table 1 summarizes the resultant mean estimated values predicted for \( R \), \( C \), and \( \kappa \) from model II for various grating spatial frequencies. We first note the large variation in the parameter estimations. The values of \( \kappa \) found with model II were larger by an order of magnitude than those obtained with model I, whereas the \( C \) values were smaller by \( \sim 25\% \). We note the unphysical negative values of \( R \approx -1.75 \) found with model II, which are comparable with a small positive \( R \approx +0.0223 \) value estimated with model I.

Because \( R = D_0K^2/F_0 \) and \( K^2 \) is a function of the spatial frequency, the value of \( R \) should increase as the spatial frequency increases. Furthermore, \( C \) and \( \kappa \) are not functions of spatial frequency, so they should remain constant as the grating period changes. From Table 1 it can be seen that they do not appear to be constant when the growth curves are fitted by the use of polynomials. For these reasons this fitting technique was deemed unsatisfactory, and we decided to proceed by fitting the full curves, using the two-harmonic analytic formulas.

B. Two-Harmonic Analytic Function Fitting Technique

To fit the growth curves we used the model II expression for \( n_1^1 \) [Eqs. (29)]. By varying the values of \( R \) and \( S \) while keeping \( C \) and \( \kappa \) constant we obtained good agreement between experimental and theoretical curves. The fixed values assigned to \( \kappa \) and \( C \) were 0.035 (m⁴/W)s and \( 1.9 \times 10^{-5} \) L/mol, respectively. We arrived at these values through manual iteration, using as a starting point the values predicted from the polynomial fitting technique described in Subsection 7.A.

The growth of unslanted transmission gratings with different grating periods was monitored during exposure, and Fresnel reflection–corrected estimates of the resulting refractive-index modulation were extracted. A model II analytic expression fitted to such a growth curve is shown in Fig. 6. The solid curve represents the theoretical fit achieved by use of the parameters shown; every tenth experimental data point is shown. The error bars represent the estimated uncertainty that is due to the acquisition system. As can be seen, although they are not exact, the fits are reasonable. Table 2 lists the values of \( S \) and \( R \) that gave the best fits with the analytic two-harmonic model II curve-fitting procedure. Note that the value of \( R \) increases as the spatial frequency is increased, while the values of \( C \) and \( \kappa \) are held fixed. It can also be seen that, as would be expected, \( S \) decreases as the spatial frequency increases and the grating period, \( \lambda \), decreases.

Table 3 summarizes the trend in the value of \( R \) as the spatial frequency increases. The \( R \) values are found to follow the \( R \approx 1/(I_0^{1/2} \lambda^2) \) relationship predicted in the literature. Furthermore, it was found that the experimental \( R \) values estimated with model II follow this theoretical relationship more closely than those found with model I. The diffusion coefficient was also estimated. For model II, \( D = 3.31 \times 10^{-14} \) cm²/s, which is almost the same as the value predicted by model I. It was found that \( R \) could be varied by as much as 15% and a good fit to experimental data still obtained. The quality of the fit is much more sensitive to the choice of values of \( S \), which can be varied only by as much as 5% before the quality of the fit deteriorates markedly. This varia-
tion in $S$ is equivalent to a 32.5% variation in $\sigma$. Outside these ranges of values of $S$ and $R$ the fits to the growth curves deteriorate, as does the agreement with the theoretical relationships.

### 8. CONCLUSION

In this paper we have generalized a nonlocal model, which we term model I,\(^8\) to take into account any form of dependence of polymerization rate on intensity and used that generalized model to derive model II.\(^8\) This generalization has been shown to be necessary\(^{12,13}\) and related to fundamental physical parameters such as the quantum yield $\Phi$ within the photopolymer. The resultant numerical predictions of the four-harmonic models I and II have been compared. They most nearly agree when $R$ is large ($R > 1$) and $S \rightarrow 1$ (local medium).

Two-harmonic analytic expressions based on model II have been presented and compared with the corresponding four-harmonic numerical predictions. The two-harmonic expressions have been shown to be accurate to within 5% for $R > 1$ (and all $S$ values) and for $S < 0.2$ (and all $R$ values). This range of validity should include many practical holographic recording materials.

Using the model II two-harmonic curve-fitting technique with $R \times 1/(I_0^{1/2}\Lambda^2)$ and $\sqrt{\sigma^2} = 75.9 \pm 25$ nm, we achieved satisfactory fits to experimental growth curves for an acrylamide-based photopolymer material with estimated values of $\kappa = 0.053$ (m/W\(^{1/2}\))/s and $C = 1.9 \times 10^{-5}$ L/mol. Diffusion coefficient $D$ was calculated to be approximately $3.31 \times 10^{-14}$ cm\(^2\)/s. The validity of these parameter estimates was discussed. It has been shown that the $R$ values estimated with model II follow the theoretically expected behavior more closely than those found with model I.

The nonnormalized value for the nonlocal variance, $\sqrt{\sigma^2} = 75.9 \pm 25$ nm, can be placed in the context of the length scales associated with general polymer chains. The length of a C–C bond is $\sim 0.15$ nm, and the carbon atom is $\sim 0.1$ nm long. Therefore a repeat unit in a polyacrylamide chain of length $\sim 0.5$ nm would not be unexpected. Given that coiled polymer chains may contain from hundreds to tens of thousands of repeat units, the lengths predicted by this model seem physically reasonable.

Although the analytic curve-fitting technique provides quite encouraging results, serious limitations of the method must be noted. The error bars associated with the experimental data and the resultant possible ranges of the estimated values of $S$ and $R$ indicate the crudeness of the estimates made. An improvement in the quality of the data and the use of an automated least-squares fitting procedure may overcome some of these difficulties.

Furthermore it was noted during application of the fitting procedures that the value for $\kappa$ estimated from the analytic curve-fitting technique (Subsection 7.B) was an order of magnitude smaller than that estimated with the polynomial curve-fitting method (Subsection 7.A). The polynomial fits were carried out close to $\xi = 0$, and the analytic technique was used to fit the whole growth curve. This would suggest that the rate of polymerization is greater at the start of the exposure than at the end. Such a variable rate of polymerization is not allowed for in any of our models; however, it seems possible that such a rate may occur because at the start of polymerization there is much more monomer available to be polymerized.

Models were recently developed that describe photopolymer systems with more than one diffusing molecule (multicomponent systems).\(^{26}\) We are currently developing coupled multiharmonic models, that incorporate these ideas. We are also exploring the affects of generalized nonlocal material response functions.\(^{27}\) We would also bring to the reader’s attention the recent Ref. 28.

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### REFERENCES


### Table 3. Model II Changes in $R$ for Increasing Spatial Frequency

<table>
<thead>
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<th>Spatial Frequency (lines mm(^{-1}))</th>
<th>$I_0$ (mW)</th>
<th>$1/(I_0^{1/2} \times \Lambda^2) \times 10^{-11}$</th>
<th>$R/R_{1000}$ (Experiment)</th>
<th>$R_{1000}^2 \times \Lambda_{1000}^5$</th>
<th>$I_0^{1/2} \times \Lambda_{1000}^5$ (Theory)</th>
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